

# Ultrafast High-Repetition-Rate Waveguide Lasers

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**Abstract**—We report on progress in the field of integrated mode-locked waveguide lasers with an emphasis on compact monolithic designs producing picosecond and femtosecond optical pulses at multi-GHz repetition rates.

**Index Terms**—Integrated Optics, Lasers, Ultrafast Optics, Waveguides.

## I. INTRODUCTION

COMPACT sources of picosecond or femtosecond optical pulses with high repetition rates have applications in a diverse range of areas including communications [1], frequency-comb spectroscopy [2,3], optical sampling [4], and nonlinear microscopy [5]. Semiconductor lasers, in either edge-emitting [6] or surface-emitting [7] formats, mode-locked fiber lasers [8], micro-ring resonators [9], and carefully engineered bulk laser systems [10] can all offer routes to sources of short pulses at multi-GHz repetition rates offering a range of attributes in terms of maximum repetition rate, average power, pulse duration, and stability. Within this field there is growing interest in waveguide lasers that can afford the potential for monolithic short cavities with fundamental mode-locked repetition rates of  $\sim 10$  GHz [11], integrated components for passive mode-locking [12] and dispersion control [13], and gain media compatible with the generation of

femtosecond pulses [14] and high average powers [15,16]. The waveguide geometry also lends itself to integrated diode-pumping [17], spatial mode control [18] and efficient heat removal [19].

In this paper we will review the progress to date in the area of ultrafast waveguide lasers. We will discuss potential gain media for such systems, the fabrication techniques that can be used to make suitable low-loss devices, the integration of components for pulse production and dispersion control, and the results achieved so far with high-repetition rate integrated waveguide laser systems.

## II. ULTRAFAST WAVEGUIDE GAIN MEDIA

The wide variety of integrated-optics fabrication techniques has allowed low-propagation-loss waveguides to be produced in a broad range of gain media. Many of these media have been shown to be suitable for the production of ultrashort pulses.

### A. Rare-Earth-Doped Glasses

Low-loss ( $\sim 0.1$  dB/cm) channel and planar waveguides are commonly produced in rare-earth-doped glasses via ion exchange. In a typical process, a sodium-rich glass is placed in a salt melt containing potassium and/or silver ions that diffuse into the glass in exchange for sodium ions, leading to a local change in refractive index that can be used to form optical waveguides. Appropriate photolithographical patterning techniques permit the fabrication of both planar (1-dimensional confinement) and single-mode channel devices. The glasses used for exchange can be compatible with relatively high rare-earth doping levels for short devices and hence high fundamental repetition rates when mode-locked as monolithic devices. Rare-earth dopant emission spectra in glasses are also typically broader than in crystal hosts, allowing the generation of ultrashort pulses. Ion-exchanged glass waveguide lasers based on  $\text{Nd}^{3+}$ ,  $\text{Yb}^{3+}$ ,  $\text{Er}^{3+}$ , and  $\text{Tm}^{3+}$  [20-23] have all been demonstrated. Other techniques for glass waveguide laser production include flame-hydrolysis deposition (FHD) [24], rf sputtering [25] and laser inscription [26].

### B. $\text{Ti}^{3+}$ -Doped Sapphire

Titanium-doped sapphire is the most prominent bulk laser system for broad tunability and the generation of ultrashort pulses. Typically, it requires pumping by argon-ion lasers or frequency-doubled diode-pumped solid-state lasers and has a

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relatively large threshold power requirement, although recent advances have paved the way for directly diode-pumped systems to be produced [27]. Consequently, there has been significant interest in creating low-loss optical waveguides in this material to deliver compact low-threshold devices. Fabrication techniques for sapphire planar waveguides include pulsed laser deposition (PLD), ion in-diffusion, ion-implantation, and laser inscription [28-31] and several waveguide laser demonstrations have been made [14, 32]. However, to date propagation losses remain relatively high, typically at the  $\sim 1$  dB/cm level.

### C. Rare-Earth-Doped Potassium Double-Tungstates

The potassium double-tungstates,  $\text{KGd}(\text{WO}_4)_2$  (KGW),  $\text{KY}(\text{WO}_4)_2$  (KYW), and  $\text{KLu}(\text{WO}_4)_2$  (KLW), offer a combination of broad absorption and emission spectra with large absorption and emission cross-sections when doped with rare-earth ions, as well as the possibility of high doping levels. They therefore represent strong competitors to Ti:sapphire as bulk gain media for ultrashort laser systems that allow simple diode-pumped devices. Optical waveguides have been fabricated in these materials by a number of different techniques including PLD, ion-implantation, laser inscription, and liquid-phase epitaxy (LPE) [33-36]. Highly efficient  $\text{Yb}^{3+}$  and  $\text{Tm}^{3+}$  doped waveguide lasers have been demonstrated based on LPE growth [37,38], which can deliver low-loss waveguides and high-optical-confinement by co-doping with Gd [39].

### D. Rare-Earth-Doped Sesquioxides

Yb-doped sesquioxides  $\text{Y}_2\text{O}_3$  (yttria),  $\text{Sc}_2\text{O}_3$  (scandia), and  $\text{Lu}_2\text{O}_3$  (lutetia) have higher thermal conductivity, broader emission spectra, and stronger absorption than the ubiquitous  $\text{Y}_3\text{Al}_5\text{O}_{12}$  (YAG), making them a good choice for high-power ultrafast bulk lasers [40]. Optical waveguides have been fabricated in sesquioxides (or mixed sesquioxides) by PLD, atomic layer deposition, and thermal evaporation [41-43], and waveguide laser operation has been demonstrated in  $\text{Nd}^{3+}$ ,  $\text{Yb}^{3+}$ , and  $\text{Tm}^{3+}$  doped crystals [44-46], with propagation losses as low as  $\sim 1$  dB/cm.

### E. Rare-Earth-Doped Garnets

The widespread prominence of garnets and especially YAG as gain host crystals for bulk lasers has led to a wide interest in creating waveguides in such a material. Successful fabrication techniques have included ion implantation, LPE, PLD, laser inscription, and contact bonding [47-51]. In particular, contact bonding is especially interesting as it allows dissimilar materials to be bonded together to produce low-loss ( $\sim 0.1$  dB/cm) multi-layered structures with high numerical apertures, forming double-clad and large-mode-area structures and allowing high-power diode-pumped operation in a similar fashion to fiber lasers [52]. While emission spectra tend to be narrower than in glasses, double-tungstates, or sesquioxides, femtosecond pulses can be generated at high powers in bulk Yb:YAG lasers [53].

### F. Chromium-Doped Gain Media

$\text{Cr}^{2+}$ ,  $\text{Cr}^{3+}$ , and  $\text{Cr}^{4+}$  doped gain media offer the potential for short-pulse devices across a broad spectral range from the

near- to mid-IR. Notably, waveguide lasers have been demonstrated in  $\text{Cr}^{2+}:\text{ZnSe}$ , fabricated by PLD and laser inscription [54,55], in  $\text{Cr}^{2+}:\text{ZnS}$  fabricated by laser inscription [56], and in  $\text{Cr}^{4+}:\text{YAG}$  [57], although the latter is in a crystal fiber rather than a planar geometry.

### G. Rare-Earth-Doped Lithium Niobate

There are a number of low-loss ( $\sim 0.1$  dB/cm) waveguide fabrication techniques, such as proton-exchange and Ti-indiffusion, which have been developed for  $\text{LiNbO}_3$  due to exceptional electro-optical, acousto-optical, and nonlinear-optical properties that make it an excellent platform for integrated optics. It is also readily doped with rare-earth ions by thermal indiffusion and numerous waveguide lasers have been demonstrated based on  $\text{Nd}^{3+}$ ,  $\text{Yb}^{3+}$ ,  $\text{Er}^{3+}$ , and  $\text{Tm}^{3+}$  [58-61].

## III. INTEGRATED SWITCHING ELEMENTS AND DISPERSION CONTROL

While both active and passive techniques have been used in integrated formats to produce pulsed waveguide laser systems, passive techniques tend to be predominant due to their ease of integration and production of shorter pulses. Actively mode-locked lasers have been demonstrated in  $\text{Er}:\text{LiNbO}_3$  waveguide lasers by depositing electrodes on the surface of the crystal to form an integrated travelling-wave phase modulator for frequency-modulated mode-locking [62]. Active mode-locking has also been achieved in an  $\text{Er}^{3+}$ -doped silica waveguide by integration with a  $\text{LiNbO}_3$  modulator via fiber-pigtails [63].

Passive mode-locking of waveguide laser systems has been demonstrated with semiconductor saturable absorbing mirrors (SESAMs), carbon nanotubes (CNTs), and graphene, [64-66], and as soliton fiber ring lasers with a planar gain element [67]. Some of these demonstrations are based on extended free-space or fiber-pigtailed cavities to enable fiber-coupled mode-locking components [65] or active control for reduction of timing jitter [68]. However, to achieve the highest fundamental repetition rates, short monolithic or quasi-monolithic cavities are desirable. This can be achieved by end-butting the mode-locking element (e.g. SESAM or graphene-coated mirror) as part of the waveguide laser cavity or by evanescent-field coupling to layers deposited on the waveguide upper surface (e.g. CNT or graphene layers). The latter technique has the better potential for power-scaled output by avoiding the high intensities associated with in-line intra-cavity elements, although it may be more difficult to achieve sufficient fluence to saturate the absorption.

The use of monolithic cavities also demands integrated dispersion control if the shortest mode-locked pulses are to be achieved. To date, this has been realized by using material in the waveguide structure with specific dispersion properties [69] and through introducing a small gap between the end-face of the waveguide and the cavity mirror to form an effective Gires-Tournois (GT) interferometer [70]. The first method has similar disadvantages to non-monolithic cavities in that the cavity length is significantly increased leading to lowered

fundamental pulse repetition rates, whereas the second is only quasi-monolithic and is prone to instability due to the need to accurately control the gap between the end-face and the cavity mirror. Future solutions could include mirrors or SESAMs with designer dispersion properties or similarly designed integrated feedback gratings [72, 73, 13]. Figure 1 shows a schematic view of a waveguide with various options for integrated passive mode-locking and dispersion control.

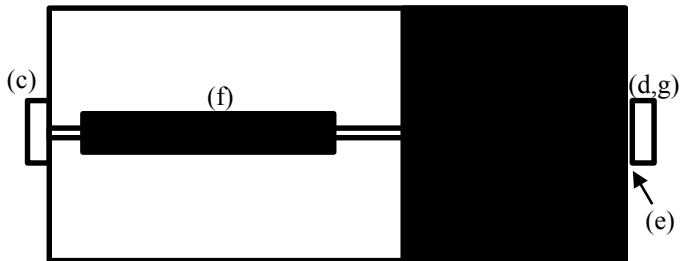


Fig. 1. Schematic channel waveguide structure top-view showing options for integrated dispersion compensation: (a) Material dispersion (b) chirped Bragg reflector (c) dispersion designed SESAM (d) chirped mirror (e) GT interferometer gap; and integrated mode-locking: (c) SESAM (f) graphene or CNT overlayer (g) graphene-coated mirror

Pulse repetition rates greater than the fundamental rate set by the cavity length may be obtained in an integrated fashion via harmonic mode-locking [73] or pulse multiplication elements [74]. However, the former can suffer from supermode-competition noise requiring integrated stabilization [75] and both add to the overall complexity of the device.

#### IV. HIGH-REPETITION RATE INTEGRATED MODE-LOCKED WAVEGUIDE LASERS

In this section we will review experimental progress to date in ultrafast high-repetition rate lasers that exploit an integrated monolithic, or quasi-monolithic, geometry and operate in

TABLE I  
ULTRAFAST HIGH-REPETITION-RATE INTEGRATED WAVEGUIDE LASERS

System	Method	PRF/GHz	$\tau_{FWHM}$ /ps	Regime
Er:LiNbO <sub>3</sub> [75]	Active Harmonic FM	10	5.7	CWML
Er:Silicate Glass [63]	Active Harmonic AM	6.3	5.2	CWML
Er:Silicate Glass [69]	Passive SESAM	0.4	0.44	CWML
Yb:Phosphate Glass [11]	Passive SESAM	15.2	0.81	CWML
Yb,Er:Phosphate Glass [12,76]	Passive SESAM	6.8	5.4	CWML
	Passive Graphene	6.8	$\geq 6.0$	QSML
Yb:Bismuthate Glass [66]	Passive Graphene	1.5	1.06	QSML
Tm:YAG Ceramic [77]	Passive Graphene	7.8	<100	QSML

PRF = Pulse Repetition Frequency,  $\tau_{FWHM}$  = Full-Width Half-Maximum Pulse Duration

either a Q-switched and mode-locked (QML) or a continuous-wave mode-locked (CWML) regime. Table 1 summarizes the results obtained to date.

##### A. Continuous-Wave Mode-Locked Waveguide Lasers

A fully integrated mode-locked waveguide laser operating at GHz repetition rates was demonstrated as early as 1993 by Suche et al. [62]. This device was based on a LiNbO<sub>3</sub> substrate, with the rare-earth doping and the waveguides fabricated by thermal indiffusion of erbium and titanium respectively. The laser cavity was formed by e-beam evaporation of alternating SiO<sub>2</sub> and TiO<sub>2</sub> layers to form mirrors on the end-faces of the waveguide. Active mode-locking at the fundamental repetition rate of 1.441 GHz was obtained with an integrated travelling-wave phase modulator fabricated by evaporating and photolithographically patterning gold electrodes onto the surface of the LiNbO<sub>3</sub> substrate. These devices were developed further, pushing to higher repetition rates through harmonic mode-locking [73] and incorporating passive low-finesse coupled cavities for supermode stabilization and pulse repetition rate multiplication [75]. Repetition rates as high as 10 GHz were achieved for near-bandwidth-limited 5.7 ps Gaussian pulses, at a wavelength of 1562 nm and output powers of  $\sim 8$  mW.

Integrated mode-locked waveguide lasers have also been constructed by pigtailed the various elements (waveguide gain medium, mode-locking element, output coupling, etc.) of the laser together using optical fibers such that free-space propagation is eliminated. Della Valle et al. [65] have incorporated a fiber-pigtailed CNT-polymer saturable absorber into a ring cavity with a femtosecond laser inscribed Yb,Er-doped phosphate glass waveguide gain medium. 1.8-ps near-bandwidth-limited sech<sup>2</sup> pulses were obtained, but the rather long fiber-pigtailed cavity led to repetition rates of just 16.74 MHz. It should be noted that a fiber-pigtailed Yb,Er-doped phosphate glass waveguide laser fabricated by ion exchange [67] has been mode-locked by exploiting nonlinear polarization rotation to achieve pulse durations as short as 116 fs but this involved the use of non-integrated free-space elements. A high-repetition-rate mode-locked fiber-pigtailed waveguide laser system was demonstrated by Kawanishi et al. [63], again employing harmonic active mode-locking. In this case a LiNbO<sub>3</sub> amplitude modulator was coupled to an Er-doped silica waveguide and 5.2 ps pulses were obtained at repetition rates of 6.3 GHz with average powers of 0.5 mW.

The first fully integrated passively mode-locked waveguide laser was demonstrated by Byun et al. [69]. The gain medium was a 5-cm-long Er-doped alumino-silicate waveguide fabricated by radio-frequency (RF) sputtering onto an oxidized silicon substrate, with etched channels and a silica upper cladding. The waveguide chip had a number of further integrated features, as shown schematically in Fig.2, including a 20-cm section of phosphorous-doped silica waveguide to control the overall group-velocity dispersion, a loop mirror to provide output coupling, a butted SESAM, an on-chip pump-coupling waveguide, and fiber-coupled pump delivery to the

chip. The looped mirror configuration allowed the relatively long total cavity length (25 cm) to be fabricated on a chip of just a few square centimeters. The device delivered pulses as short as 440 fs although the relatively long cavity length required for this method of dispersion compensation limited the repetition rate to 394 MHz. It was also shown that the monolithic design leads to low-timing-jitter operation, with a 10 kHz to 20 MHz integrated value of 24 fs.

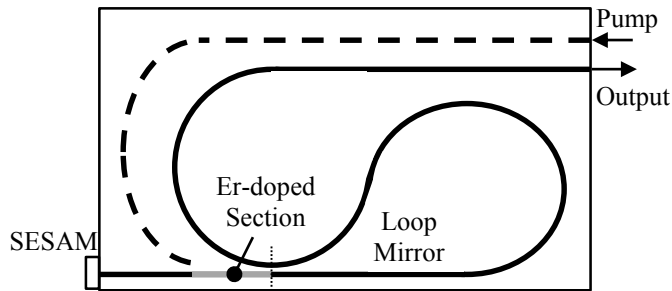


Fig. 2. Fully integrated, passively mode-locked waveguide laser schematic design [69].

Higher output powers and repetition rates were demonstrated within our group [70, 11] for a series of passively mode-locked waveguide lasers, but in a less sophisticated configuration where a SESAM and bulk output coupling mirror were simply butted to the end-faces of the waveguide to form the laser cavity. The waveguides used in these experiments were fabricated by  $K^+/Ag^+$  ion exchange in a 12 wt.% Yb-doped IOG-1 phosphate glass (Schott Glass Technologies Inc.) creating an index change of  $6.6 \times 10^{-3}$  and a diffusion depth of  $\sim 13.8 \mu\text{m}$ . Figure 3 shows a typical configuration for diode-laser pumped experiments, where a single-mode fiber-coupled diode pump laser is coupled into the waveguide with free-space optics.

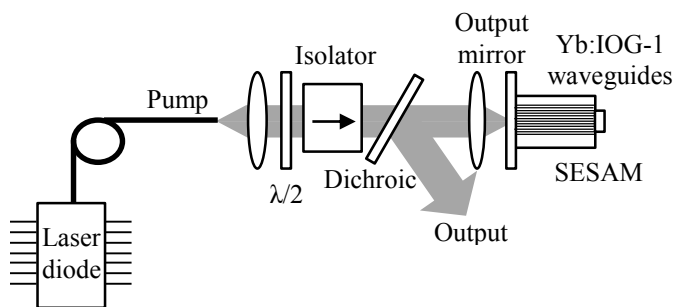


Fig. 3. Free-space diode-pumping arrangement for a passively mode-locked waveguide laser [70].

To obtain mode-locking with this set up it was found that it was necessary to introduce a small gap ( $\sim 10 \mu\text{m}$ ) between the end-face of the waveguide and the SESAM (or output mirror). We believe this gap introduced sufficient negative group-velocity dispersion, via a GT interferometer effect, to allow soliton formation, as discussed later. Figure 4 shows the output versus input characteristics for a 20-mm-long sample,

using a 4%-transmission output coupler and a SESAM (Batop GmbH) with an initial reflectivity of 99.3 % at 1050 nm, a 0.4-% modulation depth, a saturation fluence of  $90 \mu\text{J}/\text{cm}^2$  and a relaxation time of 0.5 ps.

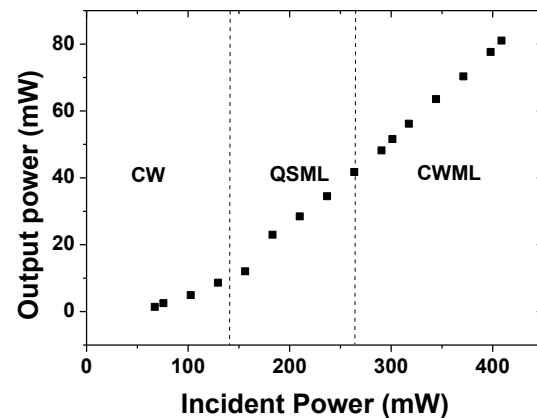


Fig. 4. Output power characteristics for a passively mode-locked Yb-doped glass waveguide laser [70].

As is commonly observed for such operation, continuous-wave (CW) output was obtained near the lasing threshold, followed by a QSML regime, and finally the desired CWML regime. At higher powers pulse break-up was observed. The intensity autocorrelation and corresponding optical spectrum of the pulses at the maximum stable output power of 81 mW are shown in Fig. 5. A  $\text{sech}^2$  fit suggested a pulse-duration of 800 fs and a time-bandwidth product of 0.46.

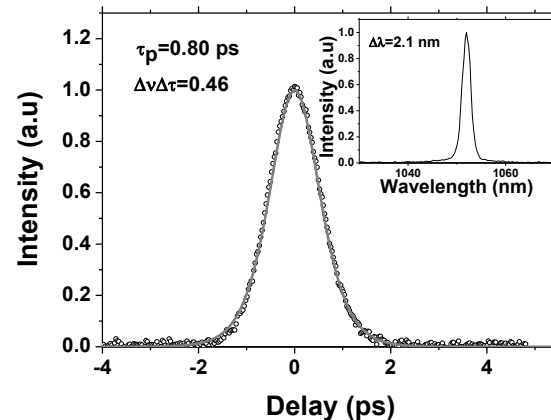


Fig. 5. Autocorrelation trace for a passively mode-locked Yb-doped glass waveguide laser (grey line is a  $\text{sech}^2$  fit). Inset shows corresponding optical spectrum [70].

The RF spectrum of the pulse train was measured with a fast InGaAs photodiode (11 GHz bandwidth) and an RF spectrum analyzer, with the results shown in Fig. 6. A clean peak was observed at the fundamental pulse repetition frequency of 4.926 GHz, with a 60-dB signal-to-noise ratio and no observable side-peaks, indicating excellent pulse-to-pulse stability and a lack of Q-switching instabilities. The wide-span RF-spectrum measurements (inset) indicated

single-pulse operation.

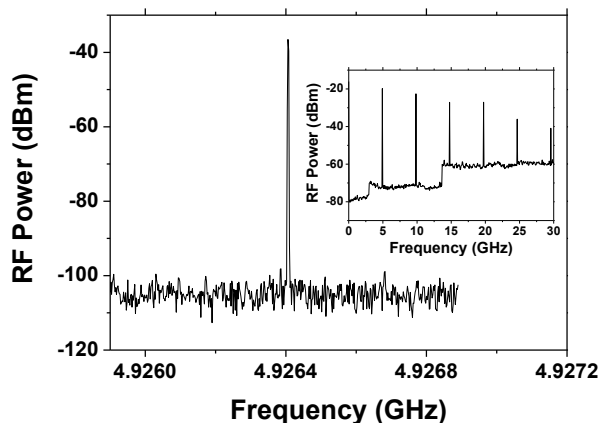


Fig. 6. RF spectrum showing the fundamental repetition frequency of a passively mode-locked Yb-doped glass waveguide laser measured with a resolution bandwidth of 1 kHz. The inset shows the broader span spectrum at a resolution bandwidth of 1 MHz. The stepwise increase observed in the baseline is an instrument artefact when the RF analyser is operated over large frequency ranges. [70].

We attributed the observed femtosecond pulses to a soliton formation mechanism, where the phase shift due to self-phase modulation (SPM) in the waveguides is balanced by negative group velocity dispersion (GVD) originating from an equivalent GT interferometer due to the micron-sized gap, as expressed below [78],

$$\frac{\delta E_p}{4\tau} = \frac{|D|}{\tau^2} \quad (1)$$

where  $\delta$  is the SPM nonlinear coefficient,  $E_p$  is the intracavity pulse energy,  $D$  is the GVD, and  $\tau$  is related to the full-width-half-maximum pulse duration by  $\tau_{FWHM} = 1.76 \tau$ . This led to an estimate of the net GVD of  $\sim -4000 \text{ fs}^2$ . Such a dispersion would be available from the equivalent GT interferometer structure [79] but there is a very sensitive variation in its value, crucially dependent upon the gap size, which would require precise control to sub- $\mu\text{m}$  precision if this method was to be applied in any practical devices. Further evidence that a soliton mode-locking regime had been achieved came from the low experimental value of the critical pulse energy required to avoid Q-switching instabilities ( $\sim 0.2 \text{ nJ}$ ) compared to that predicted without soliton effects [10] ( $\sim 3.6 \text{ nJ}$ ). Such a large reduction in the critical energy is consistent with a soliton mode-locking regime [80].

Higher repetition rates, up to 15.2 GHz, with similar pulse durations, bandwidths, and output powers, were obtained by using waveguides of different lengths (down to 6.5 mm) [11]. The limit on repetition rate will come about partly due to the need to use a length sufficient to absorb the pump light and partly due to the need to reach critical pulse energies within the cavity to avoid the QSML regime. Similar techniques were also employed to mode-lock a Yb,Er-doped IOG-1 phosphate glass producing pulses of  $\sim 2$  to 5 ps in duration, at

repetition rates up to 6.8 GHz and output powers of 30 mW [76]. A fine tuning of the repetition rate over  $\sim 1 \text{ MHz}$ , attributed to thermal expansion, was also demonstrated as the pump power was varied.

### B. Q-Switched and Mode-Locked Waveguide Lasers

Passively mode-locked lasers will operate in the QSML regime if they have not reached the critical steady-state intracavity pulse energy required for CWML operation. However, the higher mode-locked pulse energies and intensities associated with Q-switching could have potential applications in areas such as nonlinear frequency conversion and materials processing, although care must be taken to ensure that the energy contained within the Q-switched envelope does not cause damage to the mode-locking element when combined with the small spot sizes typical for single-mode waveguides.

QSML operation of a laser-inscribed Yb-doped bismuthate glass waveguide laser was achieved by Mary et al. [66] in a similar configuration to that shown in Fig. 3 except that a graphene-coated output coupling mirror was used instead of a SESAM. The 45-nm-thick graphene film was fabricated on the output coupling mirror from a filtered dispersion of graphite flakes. Raman spectroscopy of the dispersion and graphene film showed that although significant D and D' bands were present, indicating multi-layered flakes, they were electronically decoupled and behaved as a collection of single layers. The final film is estimated to contain  $\sim 40$  layers, has a saturation fluence of  $10.2 \mu\text{J}/\text{cm}^2$ , a 17.6-% modulation depth, and a 30-% non-saturable loss. A high average output power of 202 mW was achieved with a slope efficiency of 48 %, delivering near-transform-limited 1.06-ps-duration  $\text{sech}^2$  pulses at a 1.5-GHz repetition rate within a Q-switched envelope that reached repetition rates of 0.95 MHz.

A similar experiment was implemented for a laser-inscribed ceramic Tm-doped YAG waveguide laser by Ren et al. [77] to achieve QSML operation at 1943 nm. The nonlinearity of the graphene layer was measured with a 100-fs optical parametric amplifier at  $2 \mu\text{m}$  giving a  $59\text{-}\mu\text{J}/\text{cm}^2$  saturation fluence and a modulation depth of 8.4 %. A maximum output power of 6.5 mW was obtained at a slope efficiency of 2 %. The Q-switched envelope had a repetition rate up to 684 kHz and contained mode-locked pulses at a fundamental repetition rate of 7.8 GHz.

In collaboration with researchers at the University of Pittsburgh, we have also demonstrated QSML operation of waveguide lasers based on graphene-coated mirrors [12]. Ion-exchanged Yb,Er-doped IOG-1 phosphate glass was used as the waveguide gain medium, cut and polished to a length of 14.5 mm. In this case the graphene layer, fabricated at Pittsburgh, was grown by atmospheric pressure chemical vapor deposition (APCVD) on large-domain ultra-flat copper substrates [81] and then transferred onto a 2-%-transmission (at 1057 nm) output coupling mirror. Figure 7 shows the Raman spectrum of the graphene-coated mirror. The upper trace is for a similar coating on a 1535-nm mirror, which is included as the higher reflectivity of this mirror at the Raman pump wavelength of 532 nm led to a higher signal-to-noise

ratio. The observation of two distinct peaks at  $2687\text{ cm}^{-1}$  (2D) and  $1584\text{ cm}^{-1}$  (2D), along with the Lorentzian shape of the 2D peak indicated that a single-layer graphene film was present on the output mirrors. The comparison of the optical transmission spectra of the graphene-coated and uncoated mirrors indicated a  $\sim 2.3\%$  decrease in transmission due to the graphene layer, which was again consistent with a single-layer coating.

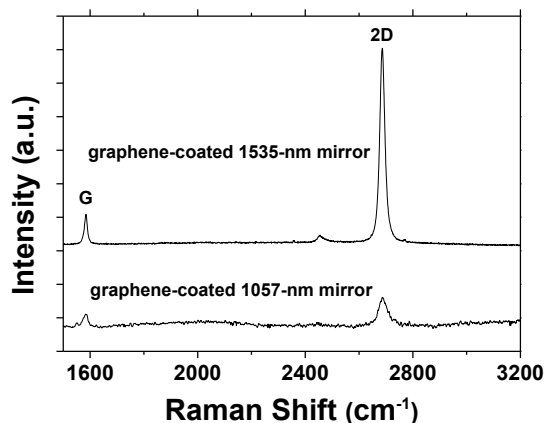


Fig. 7. Raman spectra of APCVD-grown graphene layers transferred to output coupling mirrors at 1535 nm and 1057 nm [12].

The experimental arrangement was again similar to that shown in Fig. 3, but with the SESAM replaced with the graphene-coated output coupling mirror. QSML operation was confirmed by observing the RF spectrum with a 12.5 GHz detector and an RF spectrum analyzer, as shown in Fig. 8. Mode-locking at a fundamental repetition rate of 6.8 GHz was observed and the relatively broad peak, for example in comparison to Fig. 6, is consistent with QSML operation. The Q-switched envelope and mode-locked train of pulses was observed with a 50 GHz oscilloscope, confirming QSML operation at 6.8 GHz. Q-switching repetition rates up to 526 kHz were observed and a maximum average power of 27 mW was measured at a slope efficiency of 5 %.

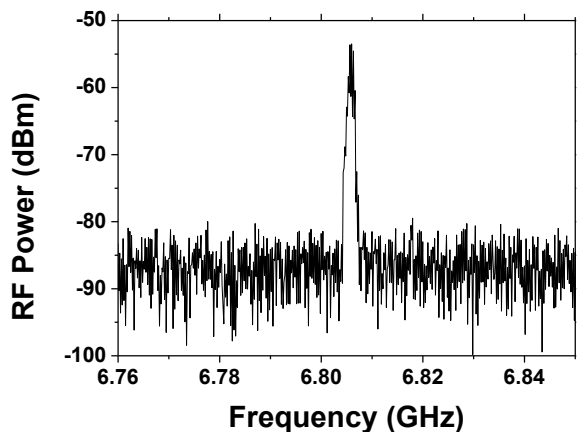


Fig. 8. RF spectrum for a Q-switched mode-locked waveguide laser [12].

## V. CONCLUSION

A range of actively and passively mode-locked waveguide lasers has been demonstrated by several research groups, delivering ultrashort pulses at multi-GHz repetition rates based on a variety of fabrication techniques, mode-locking elements, and dispersion control methods. Sub-picosecond pulses have been demonstrated by passive mode-locking, with fundamental repetition rates up to 15 GHz, limited by the need to reach critical intra-cavity pulse energies to avoid Q-switching instabilities and the need for sufficient length of gain medium to efficiently absorb the pump light. Repetition rates could be increased beyond this limit, while maintaining a compact monolithic design, by employing integrated pulse multiplication methods [74]. Currently, average output powers are also limited by the use of single-mode channel waveguides pumped by diode-lasers with maximum pump powers of  $\sim 1$  W. There are also potential damage considerations through the use of saturable absorbers against the end-faces of the waveguide. One potential method of overcoming these limitations is to use planar waveguides, with guidance in just one dimension, which would allow pumping by higher-power diode lasers [15] and exploiting evanescent-field coupling to saturable absorbers on the top surface of the waveguide. Towards this goal, we have recently demonstrated a PLD-grown Yb-doped-yttria planar waveguide pumped by a broad-area diode laser delivering up to 6 W to the waveguide. An APCVD-grown graphene layer was transferred to the upper surface of the waveguide and Q-switched output was obtained at average powers approaching 0.5 W [82]. An alternative approach to power scaling would be to adopt a master-oscillator-power-amplifier configuration, either using further high-power planar-waveguide amplifiers [83] or utilizing optical-fiber amplifiers [84].

## VI. ACKNOWLEDGEMENT

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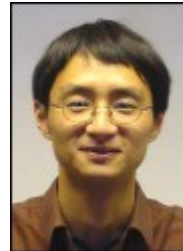
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Professor Sibbett is a Fellow of the Royal Society (London), a Fellow of the Royal Society of Edinburgh, and has Fellowships from the Institute of Physics and the Optical Society of America.



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