The Assembly-Disassembly-Organization-Reassembly Mechanism for 3D-2D-3D Transformation of Germanosilicate IWW Zeolite**

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Abstract: Hydrolysis of germanosilicate zeolites with the IWW structure shows two different outcomes depending on the composition of the starting materials. Ge-rich IWW (Si/Ge = 3.1) is disassembled into a layered material (IPC-5P), which can be reassembled into an almost pure silica IWW on treatment with diethoxydimethylsilane. Ge-poor IWW (Si/Ge = 6.4) is not completely disassembled on hydrolysis, but retains some 3D connectivity. This structure can be reassembled into IWW by incorporation of Al to fill the defects left when the Ge is removed.

In recent papers we reported a synthetic route to lamellar or 2D zeolites using the chemically selective disassembly of a 3D parent zeolite with the UTL structure and subsequently showed how new, fully tetrahedral zeolitic materials could be prepared by reassembling the 2D layers into 3D solids with new topologies. We describe this process using the ADOR (assembly-disassembly-organization-reassembly) acronym.

In this way, two zeolites were prepared: IPC-2 (which is isostructural to COK-14 with the IZA code OKO) and IPC-4 (code PCR).

We proposed that for successful application of the ADOR mechanism to other parent zeolite structures, the following structural features are required: 1) the presence of double-four-ring units (D4R) and 2) that these D4R units are preferentially occupied by Ge.

In this work, the location of the germanium atoms in the structural models was determined by careful analysis of the average interatomic distances and Fourier analysis, and showed that the only sites with significant Ge occupancy were in the D4R units. The average D4R unit in the Ge-rich sample has 6 Ge and 2 Si atoms.

The effect of hydrolysis under acidic conditions was investigated. Figure 1 shows the XRD pattern of Ge-rich IWW hydrolyzed in 0.1M HCl. The most noticeable change is a disappearance of the 111, 211, and 311 peaks, consistent with order in the c direction being reduced. The positions of peaks without contribution in the c direction are unchanged, for example, 200, 400, and 310. A new, broader peak 001, which can be attributed to the stacking of IWW layers along the c axis, is seen between 20 = 7.65 and 8.78° depending on the exact hydrolysis conditions. The hydrolysis process leads to a reduction of the interlayer distance by between 1 and 3 Å. The removal of most Ge atoms was confirmed by both chemical analysis (Si/Ge increased from 3.1 up to 45.9) and by solid state $^29$Si MAS NMR where the signal for Si Q in the original Ge atom was no longer detected. We
Hydrolysis of Ge-poor IWW, on the other hand, did not lead to such distinctive structural changes. Figure 2 compares the XRD patterns of Ge-rich IWW and Ge-poor IWW hydrolyzed under the same conditions. Hydrolyzed Ge-poor IWW does present a diffraction pattern that is distinct from the parent IWW, showing that some chemical changes have occurred. However, there are no new peaks that can be assigned to a particular ordering of the IWW layers, and—despite the changes in intensity—peaks such as the 111 and 211 remain visible. This suggests that despite the extraction of germanium from D4Rs the material has not been fully hydrolyzed into layers and that connections are still present that hold the structure together forming a defective IWW-like material (Scheme 1). The Si/Ge ratio increased from 6.4 to 50.9 and 121.5 for hydrolysis in 0.1m and 12m HCl, respectively.

To verify full separation of the layers in IPC-5P and the possibility of further manipulation, we attempted a swelling treatment using the hexadecyltrimethylammonium cation. Generally, when swelling is successful and the surfactant is stacked roughly perpendicular to the layers, we expect a shift of the 001 peak to significantly lower 2θ values. In the surfactant-treated sample, denoted IPC-5SW, the interlayer reflection 001 (in IPC-5P at 2θ = 7.8°) is shifted to lower 2θ values and now overlaps with the peak at 2θ = 7.2° (see the Supporting Information). Overall, due to the relatively small shift of the 001 reflection, we believe that the surfactant is not stacked perpendicular to the layers but horizontally between them. New reflections in the 2θ range 10–35° also appeared, which are marked with asterisks in Figure 3 (3). The positions of these peaks closely correspond to those in the parent IWW structure, as the rearrangement of the material reformed connections between the layers that were lost on hydrolysis. Since the relatively high pH used is known to promote Si/C0 bond making and breaking, this is not an entirely unexpected result. As a side effect a part of IWW layers may be dissolved and transformed into mesoporous particles of M41S type. This can be seen in the low-angle X-ray diffraction region as a broad reflection (See Figure S7.)

denote the hydrolyzed Ge-rich IWW as IPC-5P, a new lamellar material with IWW structure of the layers (Scheme 1).

Figure 1. XRD powder patterns of IWW and Ge-rich IWW (1) hydrolyzed in 0.1m HCl at room temperature for 43 h (2).

Scheme 1. Hydrolysis of IWW with different germanium content and post-synthesis treatments leading to restored IWW frameworks with different chemical composition. Aluminum atoms are shown as green dots.

Figure 2. XRD pattern of calcined Ge-rich IWW (Si/Ge 3.1) (1), its hydrolyzed forms after treatment with 0.1m HCl (2) or 12m HCl (3), and Ge-poor IWW (Si/Ge 6.4) (4) and its hydrolyzed forms after treatment with 0.1m HCl (5) or 12m HCl (6).

Figure 3. XRD patterns of Ge-rich IWW (1), hydrolyzed IPC-5P (2), surfactant-treated IPC-5SW (3), and calcined surfactant-treated-intercalated material—restored IWW (4). The asterisks mark some of the reflections newly appeared after swelling treatment.
However, when a part of the “surfactant-treated” IPC-5SW was calcined, the structure collapsed and we observe only negligible intensities in the 20 range 6–35°, suggesting that the rearrangement has not formed enough connections to lead to a stable material. The broadened peak in the low angle area may indicate the formation of mesoporous particles. To include more silicon and increase the number of interlayer connections IPC-SSW was treated with diethoxydimethylsilane (DEDMS). After calcination the material (which we denote as “restored” IWW) shows a very similar XRD pattern to the parent IWW. Removing most Ge atoms from the framework and replacing them by Si atoms decreased the unit cell size slightly. A chemical analysis confirmed the removal of Ge when Si/Ge ratio increased from 3.1 up to 73.4.

An attractive possibility is to restore IWW not with new Si from the DEDMS but with other dopant atoms, such as aluminum. Attempts to incorporate Al into defects were carried out on both hydrolyzed Ge-rich and Ge-poor samples (Figure 4) using AlCl₃ as the source of Al. The aluminated Ge-rich material displays peaks of low intensity (Si/Ge ratio increased up to 102). In contrast, the aluminated Ge-poor IWW shows the same architecture as the parent IWW. We explain the differences in aluminization of Ge-rich and Ge-poor hydrolyzed IWW as follows: When the D4R units are destroyed during the hydrolysis, as is the case of Ge-rich sample, we cannot rebuild the interlayer connections using only aluminum atoms as this would break Lowenstein’s rule. However, in the Ge-poor IWW the hydrolysis led only to defects in the D4Rs. In that case aluminum can be incorporated into the defects and thus restore a IWW structure with new chemical composition (Si/Ge 115 and Si/Al 27; Scheme 1).

The structural changes were also evaluated based on argon (Figure 5A) and nitrogen (see the Supporting Information) sorption isotherms. Parent IWW (Figure 5A (1)) is a typical microporous zeolite with BET surface area of 416 m²g⁻¹ and micropore volume of 0.169 cm³g⁻¹. The pore size distribution shown in Figure 5B,C is centered at 0.65 nm corresponding to the presence of a 12-10-8-R channel system.

After hydrolysis (to form IPC-5 after calcination) the surface area as well as micropore and total pore volume are greatly reduced (Table 1). Such behavior corresponds to

<table>
<thead>
<tr>
<th>Si/Ge (EDX)</th>
<th>BET [m²g⁻¹]</th>
<th>V_mic [cm³g⁻¹]</th>
<th>V_tot [cm³g⁻¹]</th>
</tr>
</thead>
<tbody>
<tr>
<td>Parent IWW</td>
<td>3.1</td>
<td>416.1</td>
<td>0.169</td>
</tr>
<tr>
<td>Hydrolyzed</td>
<td>45.86</td>
<td>123.6</td>
<td>0.030</td>
</tr>
<tr>
<td>Surfactant-treated</td>
<td>–</td>
<td>278.5</td>
<td>0.015</td>
</tr>
<tr>
<td>Restored IWW</td>
<td>73.38</td>
<td>509.5</td>
<td>0.146</td>
</tr>
</tbody>
</table>

a reduction in micropore volume through the destruction and removal of the D4Rs, which destroys pores parallel to the layers. However, the layers still possess 12-8-R channels going through the layers and the material still retains some micropore volume (0.030 cm$^3$).

“Swelling” of the IPC-5P structure, followed by calcination, leads to increased total surface area but this is primarily due to the formation of mesoporous particles. Hence, in the IPC-SSW isotherm (Figure 5A (3)) the continuous uptake of argon in the $p/p_0$ range 0.02–0.40 is due to the filling of small mesopores, which are also evidenced by pore size distribution analysis (a large band centered around 3 nm). After intercalation of DEDMS and restoration of the IWW structure the filling of micropores takes place in the low relative pressure region and then continues in the filling of small mesopores. The BET area increased up to 510 m$^2$ g$^{-1}$ as well as broader maxima coming from adsorption IWW hydrolysis and restoration of the calation of DEDMS and restoration of the analysis (a large band centered around 3 nm). After intercalation of Ge units, which are also evidenced by pore size distribution analysis, of various concentrations (0.1–12 m$^3$) between ambient temperature and up to 100°C. The hydrolyzed material was treated with $C_{16}$TMA surfactant and subsequently intercalated with DEDMS and calcined. The hydrolyzed solid was also directly intercalated with DEDMS or octylamine and calcined. The solid materials were characterized by X-ray powder diffraction, nitrogen and argon sorption, solid state NMR spectroscopy ($^{29}$Si MAS NMR, $^{27}$Al MAS NMR), and EDX. Details of synthesis, subsequent modifications, and spectroscopic characterization are described in the Supporting Information.

In conclusion, the location and amount of Ge in the D4Rs of IWW significantly influence the structural stability in acidic environment. The hydrolysis results for Ge-rich and Ge-poor IWW materials show two possible outcomes. Full hydrolysis of Ge-rich materials resulted in a 2D solid (IPC-5P). The layered IPC-5P was converted back to the 3D structure of IWW by incorporation of silylating agent. The restored IWW has a significantly higher Si/Ge ratio (73) than the parent IWW (3.1).

The hydrolysis of Ge-poor IWW also led to extraction of germanium from mixed Si/Ge D4R units. However, because there is much less Ge in the D4R than for the Ge-rich sample, the material preserved enough interlayer connections to retain its 3D IWW framework, albeit with significant defects. Moreover, the structural defects can be filled by incorporation of Al into defined crystallographic positions.

Our work confirms the application of the ADOR mechanism on another zeolitic structure—IWW. In contrast to zeolite UTL where we obtained two new zeolites (OKO, PCR), here the layered structure IPC-5P strongly tends to reform the original IWW framework.

**Experimental Section**

The preparation of IWW followed the procedure described in Ref. [9]. The calcined IWW samples were treated in different acids (HCl, H$\text{NO}_3$, CH$_3$COOH) of various concentrations (0.1–12 m$^3$) between ambient temperature and up to 100°C. The hydrolyzed material was treated with $C_{16}$TMA surfactant and subsequently intercalated with DEDMS and calcined. The hydrolyzed solid was also directly intercalated with DEDMS or octylamine and calcined. The solid materials were characterized by X-ray powder diffraction, nitrogen and argon sorption, solid state NMR spectroscopy ($^{29}$Si MAS NMR, $^{27}$Al MAS NMR), and EDX. Details of synthesis, subsequent modifications, and spectroscopic characterization are described in the Supporting Information.

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